

Fig. 2. A general view of the asymmetric unit of hydrated cobalt(II) 4-methyltropolonate (one non-bonded $\mathrm{H}_{2} \mathrm{O}$ molecule omitted). Broken lines denote hydrogen bonds.

The asymmetric unit (Fig. 2) consists of a dimer $(A)$ with formula $\left[\mathrm{Co}(\mathrm{MeT})_{2} \mathrm{H}_{2} \mathrm{O}\right]_{2}$, which has bridging MeT oxygen atoms and is structurally analogous to $\left[\mathrm{NiT}_{2} \mathrm{H}_{2} \mathrm{O}\right]_{2}$, and a mononuclear species $(B)$ with formula $\mathrm{Co}(\mathrm{MeT})_{2}\left(\mathrm{H}_{2} \mathrm{O}\right)_{2}$ which exhibits cis configuration of water molecules. A nonbonded water molecule is also present. Hydrogen bonds exist between coordinated water molecules and ligand $O$
atoms; both $\mathrm{H}_{2} \mathrm{O}$ molecules of $B$ take part in this interaction but only one $\mathrm{H}_{2} \mathrm{O}$ on $A$ is involved, such that each $B$ molecule is associated via three H -bonds ( $\mathrm{O}-\mathrm{O}$ distances $2.61,2.73$ and $2.75 \AA$ ) with one $A$ molecule only. It is notable that there is no interaction between separate $A$ pairs or $B$ pairs, other than that induced by normal van der Waals approaches and that many of the shorter intermolecular contacts involve methyl C atoms.

Although the mother liquor must have contained both species, the relative concentration of each, and possibly others, in solution could have been a structure-determining factor for the solid state. However, the observed constitution of the stable lattice (the crystals were not moisture sensitive) would appear necessary to facilitate H-bonding and lattice packing which, because of the methyl substituent, is possibly not so efficiently achieved when either species is absent.

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Electron-density distribution in rutile crystals. By H. Shintani, S. Sato and Y. Saito, Institute for Solid State
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The distribution of residual electron density in rutile crystals has been calculated from intensities carefully obtained by diffractometry. There is no indication that the charge densities of $\mathrm{Ti}^{4+}$ (having no $d$ electrons) are deformed in an octahedral crystal field. This result seems to support our conclusion that the aspherical charge densities of $\mathrm{Ni}^{2+}\left(3 d^{8}\right)$ in $\gamma-\mathrm{Ni}_{2} \mathrm{SiO}_{4}$ are due to $d$ electrons placed in an octahedral crystal field.

An indication of $3 d$ electrons in the $t_{2 g}$ orbitals of $\mathrm{Ni}^{2+}$ and $\mathrm{Co}^{3+}$ was observed in the final difference syntheses of $\gamma-\mathrm{Ni}_{2} \mathrm{SiO}_{4}$ and $\left[\mathrm{Co}\left(\mathrm{NH}_{3}\right)_{6}\right]\left[\mathrm{Co}(\mathrm{CN})_{6}\right]$ (Marumo, Isobe, Saito, Yagi \& Akimoto, 1974; Iwata \& Saito, 1973). In the difference maps eight small peaks were arranged at the corners of a cube around the transition metal atom, the peaks being at $0.45 \AA$ from the metal atom. To see whether or not such peaks are due to $d$ electrons, a difference synthesis of rutile, $\mathrm{TiO}_{2}$, was calculated based on carefully measured intensity data, since $\mathrm{Ti}^{4+}$ possesses no $3 d$ electrons.
All attempts to shape a crystal specimen into a sphere failed. The specimen used had the dimensions $0.08 \times 0.08 \times$ 0.09 mm . The intensities were collected on a Rigaku automated four-circle diffractometer. The experimental conditions were exactly the same as those for $\gamma-\mathrm{Ni}_{2} \mathrm{SiO}_{4}$,

Table 1. Crystal data and atomic parameters

| $\mathrm{TiO}_{2-\alpha}, \alpha=0.016(7)$ | $U=62.072 \AA^{3}$ at $26^{\circ} \mathrm{C}$ |
| :--- | :--- |
| Tetragonal, $P 4_{2} / m n m$ | $D_{m}=4.264 \mathrm{~g} \mathrm{~cm}^{-3}$ |
| $a=4.5845(1) \AA$ | $D_{x}=4.260 \mathrm{~g} \mathrm{~cm}^{-3}$ |
| $c=2.9533$ (1) $\AA$ | $Z=2$ |

O at $(0, u, u)$ with $u=0.30493$ (7)
Isotropic extinction parameter $g=0.29 \times 10^{4}$
The temperature factors are in the form:

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Table 2. Observed and calculated structure amplitudes

except that the maximum number of repetitions of a measurement was ten. Equivalent reflexions $h k l$ and $h k l$ were measured and averaged. In all, 302 independent data were collected and then corrected for Lorentz-polarization and absorption. The structure was refined by the full-matrix least-squares program LINUS (Coppens \& Hamilton, 1970). The site-occupancy parameter, $\alpha$, in the formulation $\mathrm{TiO}_{2-a}$ was also refined. The final $R$ value was reduced to 0.010 and $R w$ was 0.014 for the 302 observed reflexions. The atomic scattering factors for $\mathrm{Ti}^{4+}$ and $\mathrm{O}^{2-}$ were taken from Fukamachi (1971).
The crystal data and the final atomic parameters are listed in Table 1. The $u$ parameter of 0.30493 (7) agrees well with the previously obtained value, $0 \cdot 30479$ (10), within experimental error (Abrahams \& Bernstein, 1971). The observed and calculated structure amplitudes are compared

Fig. 1. Section of the difference Fourier function through the titanium ion and perpendicular to the $c$ axis. Contours are lines, zero contours chained. Negative contours are broken
in Table 2. The final difference syntheses were calculated and a section through the titanium ion and perpendicular to the $c$ axis is shown in Fig. 1. The resulting maps were featureless to within $0.3 \mathrm{e} \AA^{-3}$. There were no meaningful peaks around the titanium atom, indicating that the charge
density around the $\mathrm{T}^{4+}$ is spherical as expected to support our result previously obtained expect. This seems to support our result previously obtained for $\gamma-\mathrm{Ni}_{2} \mathrm{SiO}_{4}$.

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